

# Observation of Interstitial Molecular Hydrogen in Clathrate Hydrates\*\*

R. Gary Grim, Brian C. Barnes, Patrick G. Lafond, Winfred A. Kockelmann, David A. Keen, Alan K. Soper, Masaki Hiratsuka, Kenji Yasuoka, Carolyn A. Koh, and Amadeu K. Sum\*

**Abstract:** The current knowledge and description of guest molecules within clathrate hydrates only accounts for occupancy within regular polyhedral water cages. Experimental measurements and simulations, examining the *tert*-butylamine + H<sub>2</sub> + H<sub>2</sub>O hydrate system, now suggest that H<sub>2</sub> can also be incorporated within hydrate crystal structures by occupying interstitial sites, that is, locations other than the interior of regular polyhedral water cages. Specifically, H<sub>2</sub> is found within the shared heptagonal faces of the large (4<sup>3</sup>5<sup>9</sup>6<sup>2</sup>7<sup>3</sup>) cage and in cavities formed from the disruption of smaller (4<sup>4</sup>5<sup>4</sup>) water cages. The ability of H<sub>2</sub> to occupy these interstitial sites and fluctuate position in the crystal lattice demonstrates the dynamic behavior of H<sub>2</sub> in solids and reveals new insight into guest–guest and guest–host interactions in clathrate hydrates, with potential implications in increasing overall energy storage properties.

When water and low-molecular-weight compounds are mixed at high pressures and low temperatures, crystals of hydrogen-bonded water cages, known as clathrate hydrates, can form. The clathrate hydrate structure is differentiated from other solid forms of water by the incorporation of a single guest molecule inside regular polyhedral water cages, stabilizing the crystal structure.<sup>[1]</sup> Notable exceptions to this heuristic model include very small molecules such as H<sub>2</sub> and N<sub>2</sub>, which can multiply occupy larger water cages at moderate pressures.<sup>[2–4]</sup> The ability to capture compounds within a water

framework is currently an area of both scientific and practical interest for applications including energy storage, carbon sequestration, and hydrocarbon production.<sup>[5–7]</sup> Whether considering singly or multiply occupied water cages, the accepted knowledge is that the stability of the clathrate framework is dependent on the interaction between enclathrated guests and the host water lattice.<sup>[8,9]</sup> Specifically, if clathrate occupancies are reduced below a certain threshold, ice or water will be thermodynamically preferred, suggesting a critical and balanced interaction between guest and host molecules. This constraint of guest molecules inside individual water cages has been well-supported from single-crystal X-ray diffraction, neutron diffraction, and quantum chemical calculations.<sup>[2,10–15]</sup> Herein, we provide evidence from experiments and simulations on the structure VI clathrate hydrate system that suggests hydrate guests may also occupy interstitial sites within the host water lattice, which is in contrast to the conventional definition of guest molecules residing only in the interior of well-defined polyhedral water cages.

Structure VI (sVI) hydrates are the thermodynamically stable phase of *tert*-butylamine (tBuNH<sub>2</sub>) + H<sub>2</sub>O for temperatures below 271 K at 0.1 MPa. The sVI unit cell contains 12 octahedra 4<sup>4</sup>5<sup>4</sup> cages and 16 larger heptakaidecahedra 4<sup>3</sup>5<sup>9</sup>6<sup>2</sup>7<sup>3</sup> cages, forming a cubic (*a* = 18.81 Å) crystalline structure where tBuNH<sub>2</sub> occupies the large cage and the small cage remains vacant.<sup>[16]</sup> Unlike the other more common hydrate structures (I, II, H) that easily form binary systems (occupancy with two different guests), sVI has been experimentally observed with only a single guest, tBuNH<sub>2</sub>. Despite possessing the highest theoretical H<sub>2</sub> storage capacity for a clathrate hydrate structure, attempts to synthesize binary sVI hydrates with H<sub>2</sub> or CH<sub>4</sub> have shown a rapid and complete transition to structure II hydrate at temperatures greater than 243.15 K.<sup>[17–19]</sup> Herein, we show that by reducing the system temperature to 123–153 K, the transition from sVI to sII can be significantly delayed on the order of several days, allowing for the exploration of the native sVI environment preceding the structural transition to sII.

After pressurizing pre-formed sVI tBuNH<sub>2</sub> hydrate samples (9.31 mol% tBuNH<sub>2</sub>) with H<sub>2</sub> at 70 MPa and 143 K, Raman spectroscopic analysis reveals a distinct H<sub>2</sub> vibron centered at about 4136 cm<sup>−1</sup> (Figure 1A). Interestingly, this vibron lacks the spectroscopic characteristics typically found for H<sub>2</sub> in hydrates.<sup>[3]</sup> Specifically, whereas in the Raman spectra of other binary H<sub>2</sub> hydrates structures (sI, sII, sH; Figure 1B–D) each enclathrated H<sub>2</sub> environment is clearly defined by ortho/para H<sub>2</sub> peaks separated by about 6 cm<sup>−1</sup>, the new H<sub>2</sub> environment (1A) is comparatively much broader (spanning a total of about 40 cm<sup>−1</sup>) and without clear features,

[\*] R. G. Grim, Dr. B. C. Barnes, Dr. P. G. Lafond, Dr. C. A. Koh, Dr. A. K. Sum

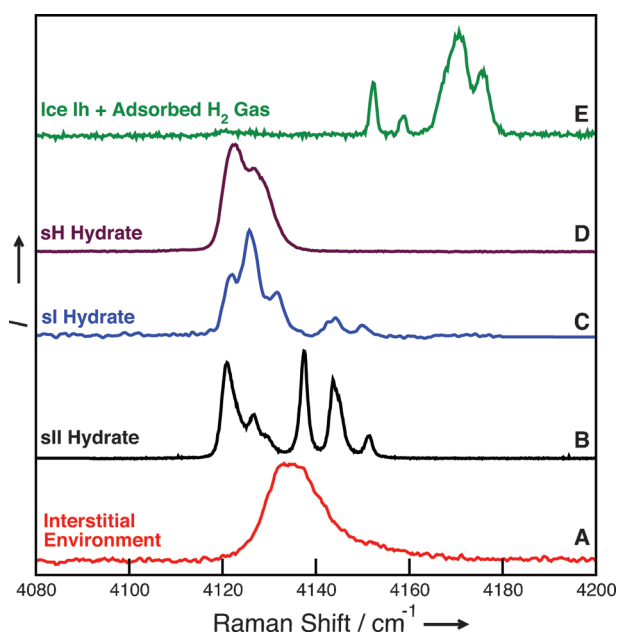
Department of Chemical and Biological Engineering  
Colorado School of Mines  
1500 Illinois Street, Golden, CO 80401 (USA)  
E-mail: asum@mines.edu

Dr. W. A. Kockelmann, Dr. D. A. Keen, Dr. A. K. Soper  
ISIS Facility, Rutherford Appleton Laboratory  
Chilton OX11 0QX, Oxon (U.K.)

M. Hiratsuka, Dr. K. Yasuoka  
Department of Mechanical Engineering, Keio University  
3-14-1 Hiyoshi, Kohoku-Ku, Yokohama 223-8522 (Japan)

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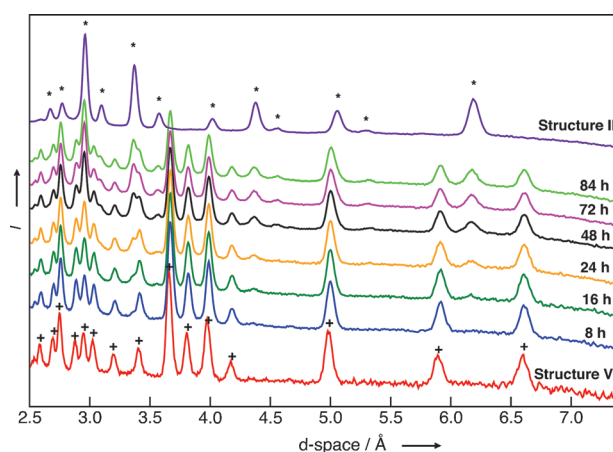
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**Figure 1.** Raman vibrational spectra of  $\text{H}_2$  in known clathrate hydrate environments of sI, sII, sH, and ice Ih (B–E), and in the proposed interstitial environment(s) in sVI (A), highlighting differences in features, position, and width. All of the Raman spectra were measured at 0.1 MPa and 83 K.

suggesting possible convolution of several environments. Additional experiments studying ortho/para  $\text{H}_2$  conversion after  $\text{LN}_2$  exposure suggest a minimum of two unique sites (Supporting Information, Figure S7). Given that sVI hydrates are thermodynamically driven to transition to sII after the addition of  $\text{H}_2$ , it is expected that only three phases would exist throughout the transition process: sVI (initial structure), sII (final structure), and possibly ice Ih as an intermediate byproduct of crystal reformation.<sup>[17,18]</sup> In comparing the observed  $\text{H}_2$  environment (1A) to a pure  $\text{H}_2$  hydrate sII sample formed at 114 MPa/263 K (1B) and  $\text{H}_2$  in ice Ih<sup>[7,20]</sup> (1E), the peak shift, shape, and width are significantly different, suggesting the observed  $\text{H}_2$  vibron cannot be associated with  $\text{H}_2$  in sII or ice Ih, but possibly a native sVI environment. As the formation process of the initial sVI hydrate sample was conducted using  $\text{H}_2\text{O}$  and  $t\text{BuNH}_2$  at the stoichiometric concentration for sVI clathrate formation ( $16t\text{BuNH}_2 + 156\text{H}_2\text{O}$ ), it is assumed that all of the large  $4^35^96^27^3$  cavities are fully occupied by a single  $t\text{BuNH}_2$  molecule. By process of elimination, this suggests only the  $4^45^4$  small cage of sVI would be available for  $\text{H}_2$  enclathration and potentially responsible for the vibron in Figure 1A.

To more accurately characterize the hydrate structure(s) present during the transition process, in situ neutron diffraction was performed at 70 MPa and 190 K over a period of 84 h in the presence of  $\text{D}_2$  (Figure 2). The structure of the initial hydrate sample was indexed as cubic  $I\bar{4}3d$  ( $a = 18.60401(2)$  Å) confirming a pure sVI phase.<sup>[16]</sup> As  $\text{D}_2$  was added to the system, the sample slowly began to transition, showing characteristic sII reflections within the first 8 h. After 84 h, approximately 28% of the initial sVI had converted into sII ( $Fd\bar{3}m$ ,  $a = 17.4806(1)$  Å).<sup>[1]</sup> To finalize the transition to sII,



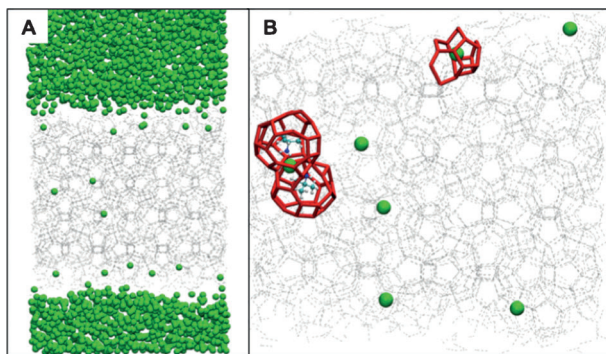
**Figure 2.** In situ neutron diffraction data from the General Materials Diffractometer (GEM; detector bank 3), covering the complete sVI-to-sII transition acquired at 70 MPa and 190 K over a range of 84 h of exposure to  $\text{D}_2$ . Diffraction peaks are denoted as sVI (+) and sII (\*).

the sample was heated to 263 K and held for 8 h, during which complete conversion to sII was observed, and was then cooled back to 190 K for the final acquisition. Throughout the entire transition experiment, only two crystal structures were detected, sVI and sII. The lack of a distinct third phase suggests that the  $\text{H}_2$  vibron observed from the Raman spectroscopic measurements (Figure 1A) could only be derived from  $\text{H}_2$  enclathration in either sVI or sII. Furthermore, when combined with the Raman analysis discussed above, the data confirm that the  $\text{H}_2$  signal is a result of enclathration into sVI. Additionally, during the neutron scattering experiment, it was observed that immediately following the initial pressurization with  $\text{D}_2$  to 70 MPa, the sVI lattice expanded from 18.6040(12) Å to 18.6645(09) Å (Supporting Information, Figure S2).

From previous work, it is known that hydrate lattice constants are affected by at least three variables: temperature, pressure, and occupancy of guests. The reported expansion was observed at constant temperature and following an increase in pressure, which acts to compress the lattice, suggesting the expansion may result from  $\text{D}_2$  insertion into sites within the sVI lattice.<sup>[4,21]</sup> To quantify the effect of  $\text{H}_2$  penetration into the hydrate lattice and its expansion, molecular simulations studying sVI unit cell dimensions as a function of guest filling were performed. These simulations showed that, as expected, the insertion of  $\text{H}_2$  molecules in the sVI lattice resulted in gradual increase in lattice constant of the crystal and with a loading of approximately 8  $\text{H}_2$  molecules per unit cell, a comparable increase in lattice constant as in the case of the neutron experiment was observed (Supporting Information, Figure S5).

To probe possible  $\text{H}_2$  occupancy within native sVI environments, a series of classical molecular dynamic (MD) simulations for the  $t\text{BuNH}_2 + \text{H}_2 + \text{H}_2\text{O}$  system was performed. Based on earlier experimental efforts studying sVI +  $\text{H}_2$  binary hydrates and our DFT calculations (Supporting Information, Table S2), it was expected that  $\text{H}_2$  would be unable to occupy any native sVI environment as the large

( $4^35^96^27^3$ ) cages were fully occupied by  $t\text{BuNH}_2$  and the small ( $4^45^4$ ) cages were shown to be too constrained for enclathration.<sup>[17,18,22]</sup> Surprisingly, the simulations revealed that  $\text{H}_2$  readily diffuses into the sVI lattice (Figure 3A), assisted



**Figure 3.** A) Simulated system consisting of 27 sVI unit cells (4212  $\text{H}_2\text{O}$  molecules with 432  $t\text{BuNH}_2$  molecules) and 2393  $\text{H}_2$  molecules at 230 K and 85 MPa. B) Snapshot from the simulation showing seven  $\text{H}_2$  molecules within the sVI lattice, highlighting the two dominant types of interstitial sites in red. Green spheres represent the center of mass of  $\text{H}_2$  (size increased for clarity), and gray dotted lines correspond to hydrogen bonds between water molecules.

primarily by the metastability of the host water network. The results show that upon entering the sVI crystal,  $\text{H}_2$  did not occupy either of the well-defined sVI water cages, but instead was localized in two interstitial sites (Figure 3B).<sup>[23,24]</sup> In one site,  $\text{H}_2$  migrates through the crystal lattice, and resides in the heptagonal-shared face of the  $4^35^96^27^3$  water cages (Figure 4A). In the other site, the inability of  $\text{H}_2$  to occupy either of the two thermodynamically stable cage types in sVI induces local disruption of the water hydrogen bond frame-

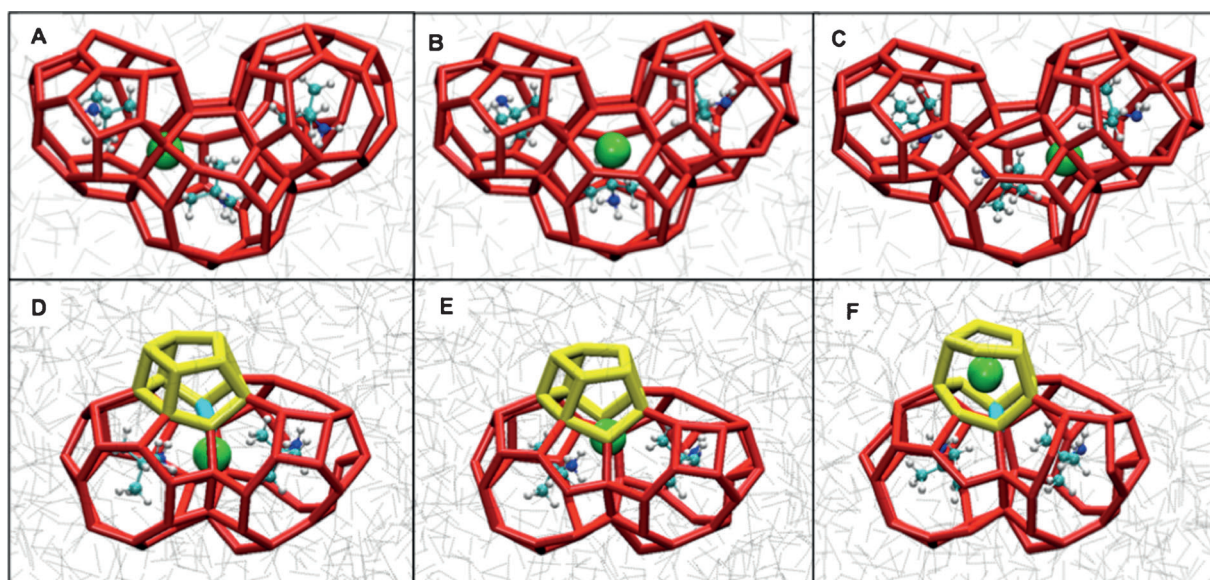
work, specifically in the  $4^45^4$  cages, resulting in cavities with disrupted hydrogen bonds that can also stabilize  $\text{H}_2$  (Figure 4B). A study on the residency times of individual  $\text{H}_2$  molecules upon diffusion into the bulk sVI structure revealed that approximately 55 % of the total time was spent in a disrupted small cage environment, 40 % inside of heptagonal shared face, and 5 % as a co-guest with  $t\text{BuNH}_2$  inside the large cage. It should be emphasized that when  $\text{H}_2$  was observed in these interstitial spaces, it was typically stabilized on the order of 10–20 ns, whereas the transition step or “hop”, from one interstitial space to another occurred quickly (< 5 ps), revealing that the timescale for the localization of  $\text{H}_2$  at interstitial sites is roughly four orders of magnitude greater than the diffusion step (see the Supporting Information for videos illustrating these processes).

Prior to this study, hydrate guest occupancies in interstitial sites had not been considered, and these results provide new insight into guest–guest and guest–host interactions, highlighting the dynamic environment of both guest and host molecules in the clathrate structure. While this finding of interstitial occupancy exists only as a metastable state during the sVI to sII transition, these results suggest that it may be possible to discover other combinations of clathrate hydrate structures plus guests to form thermodynamically stable interstitial guest occupancy, which would have significant implications in increasing overall storage capacity for a variety of applications.

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**Figure 4.** A–C) Pathway and interstitial sites occurring during the  $\text{H}_2$  (green) migration between two heptagonal faces shared among three fully occupied  $4^35^96^27^3$  cages (red). D)–F) Pathway for stabilization of  $\text{H}_2$  in cavity formed from disrupted hydrogen bonds (blue) in the small  $4^45^4$  cage (yellow).



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